# A Self-charging Concentration Cell: Theory

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Batteries are a key resource in the quest for sustainable energy. Here the theoretical basis is presented for a new type of electrochemical concentration cell that might contribute to this enterprise. The cell, which has been successfully demonstrated in the laboratory (Sheehan, *et al.*)<sup>1</sup>, incorporates a *chemically asymmetric membrane* to drive anisotropic diffusion between two solution chambers; the resulting concentration difference powers the cell. In this study, the membrane's operation is validated via three theoretical approaches: (i) traditional equilibrium thermodynamics; (ii) balancing drift and diffusion current densities; and (iii) the time-independent diffusion equation. The physical criteria for its operation are developed and its dimensionless variables identified. The cell's maximum instantaneous power density might exceed  $10^7 \text{ W/m}^3$ . Its self-charging capability should confer multiple advantages over traditional concentration cells (as well as over some voltaics), including improved thermodynamic efficiency, economy, and compactness. Commonalities with other electrochemical systems (e.g., liquid chromatography, metal corrosion, and solid state diodes) are discussed and a physical instantiation of the cell is reviewed. Recent numerical simulations corroborate its essential processes<sup>2</sup>.

Keywords: concentration cell, sustainable energy, second law of thermodynamics, electrochemical cell, rechargeability

#### I. INTRODUCTION

Electrochemical cells are hallmarks of modern civilization and are also among its oldest technologies, dating back at least to 1790 with the research of Galvani and Volta<sup>3-6</sup>. They serve in all technological strata and have become crucial to the rise of sustainable energy; their worldwide power storage capacity is currently 1-2 GW – and rising exponentially<sup>6</sup>. It is estimated that spending on galvanic (voltaic) storage systems exceeds 100 billion USD per year, while additional billions are devoted to their research and development, making them among the most intensely studied physical systems in the world.

Electrochemical cells come in several varieties. The most common, *voltaic* cells, transduce chemical energy into electrical energy; in fact, almost any chemical reaction involving the transfer of electrons can be parlayed into a battery<sup>7</sup>. Voltaics transfer electrons between *disparate* chemical species as, for example, in the hydrogenoxygen fuel cell:  $2H_2 + O_2 \longrightarrow 2H_2O$ . In contrast, a *concentration cell* generates electricity from the concentration difference between two samples of a *single* chemical species, exploiting their entropy of mixing.

Concentration cells usually provide smaller emfs than voltaics (e.g., ~ 0.1V vs. ~ 1V) and lower energy densities as well (e.g.,  $\rho_{\rm e} \sim 10^5 - 10^6 {\rm J/m^3}$  vs.  $\rho_{\rm e} \sim 10^8 - 10^9 {\rm J/m^3}$ ). Nevertheless, large scale concentration cells have found niche applications<sup>8–10</sup>. To illustrate, the chemical potential attainable in playing the salinity of seawater against that of fresh water (e.g., where a river flows into the sea) is comparable to the gravitational potential of a 200 m high dam. However, because tradi-

tional free energy sources are richer, more ubiquitous, exhibit larger emfs and energy densities, plus are usually easier and less expensive to tap, concentration cells are often overlooked. This is an undeserved oversight.

In Nature, concentration cells and gradients are ubiquitous and crucial sources of energy. For instance, salinity gradients are partially responsible for thermohaline circulation in the world's ocean, which is integral to global heat and matter transport as well as to climate stability<sup>11</sup>. Proton gradients across biological membranes store electrostatic potential energy and are converted into chemical energy, the most famous example of which is oxidative phosphorylation in mitochondrial membranes of eukaryotes, converting ADP into ATP<sup>12</sup>. Action potentials (nerve impulses) stem from the relaxation of sodium and potassium ion gradients across membranes of various so-called excitable cells (e.g., neurons, muscle cells, endocrine cells)<sup>13</sup>.

This study develops the theory of a new type of concentration cell. This asymmetric membrane concentration cell (AMCC) generates electricity identically to traditional concentration cells, but it is distinguished by the nonstandard means by which its concentration difference is established and maintained  $(\Delta[A] \equiv [A]_{high} - [A]_{low}).$ Here, [A] is given in mole/liter (i.e., molarity) or in particles/m<sup>3</sup>. Whereas in traditional cells  $\Delta[A]$  is arranged *externally* via prepared solutions, thereby incurring costs in thermodynamic work, in an AMCC it is generated *internally*, tapping thermal energy within the cell itself to promote anisotropic molecular diffusion of ions (or neutrals) through a chemically asymmetric membrane. This process entails no external thermodynamic work and it confers several potential advantages over its competitors, including improved economy, thermodynamic efficiency, and compactness. The AMCC concept can be applied to many types concentration  $cells^{8-10}$  and

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perhaps even to some types of voltaics<sup>3–5</sup>. It has been successfully demonstrated in the laboratory<sup>1</sup> and numerical simulations strongly corroborate its basic physical processes<sup>2</sup>. Either neutral or ionic species can be active solutes; here, for explanatory simplicity, we consider neutral species only.

The heart of the AMCC is a chemically asymmetric membrane that establishes an equilibrium concentration gradient. The physical and chemical assumptions leading to it are unremarkable, but the resulting effect – selfchargeability – is surprising and novel.

In this study, the membrane's operation is validated by three complementary theoretical approaches: (i) traditional equilibrium thermodynamics (§II); (ii) competition between drift and diffusion current densities (Appendix A); and (iii) solutions to the time-independent diffusion equation (Appendix B). The physical criteria and dimensionless parameters governing its operation are derived. Dimensional analysis suggests that its maximum power density could be substantial, perhaps in excess of  $10^7$ W/m<sup>3</sup>. Similarities to other electrochemical and physical systems (i.e., liquid chromatography, metal corrosion, and solid state diodes) provide additional insight into its character.

This article is organized as follows: §II develops the thermodynamics of the AMCC concentration gradient; §III discusses analogies to semiconductor systems as well as connections to metal corrosion and chromatography; and §IV summarizes the main results and proposes directions for future research. Appendices A and B describe other theoretical approaches for deriving the concentration gradient, and Appendix C reviews a physical instantiation of the AMCC<sup>1</sup>.

#### **II. METHODS**

The AMCC generates electricity from a concentration difference produced by a chemically asymmetric membrane. It consists of two subsystems: (i) the *asymmetric membrane separator* (AMS), which generates the concentration difference  $\Delta[A]$  between two liquid reservoirs; and (ii) the concentration cell that converts this  $\Delta[A]$  into electricity. The former is the focus of this study; the latter is extensively described elsewhere<sup>3-6,8-10</sup> and so will be considered only incidentally.

The AMS can be analysed from several viewpoints. In this section it is developed from a thermodynamic perspective as an array of chemically reactive microscopic tubes that generates an equilibrium gradient in solute concentration.

## A. AMS Model: Tube Array

The AMS consists of two thin liquid reservoirs separated by a chemically asymmetric membrane. Here the membrane is modeled as an array of hollow, large aspect



(a)

(b)

0

FIG. 1. Depictions of AMS membrane and species concentration profiles. (a) Cross section of full membrane modeled as an array of microscopic tubes (top) and magnification of a single tube extracted from the membrane and rotated for clarity (bottom). (b) Plot of initial distribution of binding sites  $([B(z)]_o \text{ in red})$  and resultant equilibrium concentration of solute ([A(z)] in blue) along length of tube  $(0 \le z \le L)$ . Here the distributions are portrayed as sigmoid functions. Reciprocal spatial relation between [A] and  $[B]_o$  follow from equilibrium considerations, Eqs. (2-5). The concentration differential  $(\Delta[A(z)] = [A(L)] - [A(0)])$ , powers the AMCC.

z→

L

ratio tubes bundled together lengthwise like a sheaf of wheat and filled with species A dissolved in a solvent, e.g., water, methanol, or acetone (Fig. 1a). Typical tube dimensions are in the range:  $10^{-6}$ m  $\leq L \leq 10^{-4}$ m and  $10^{-8}$ m  $\leq w \leq 10^{-6}$ m. Billions or trillions of tubes can comprise a single AMS membrane. They are assumed to be straight and uniform, thereby sidestepping issues of tube tortuosity and constriction, both of which complicate, but do not illuminate, the essential physics. The liquid reservoirs are sufficiently thin that their A concentrations are dominated by those that suffuse the membrane. The tubes are identical, therefore, to understand the physical chemistry of a single tube is to understand that of the entire membrane.

Firmly secured to the inner walls of each tube are chemical binding sites B that undergo the following reversible chemical reaction with the solute A:

$$A_{(l)} + B_{(s)} \rightleftharpoons AB_{(s)}.\tag{1}$$

Species A is dissolved in a liquid solvent (l), while species AB and B are bound surface states (s) on the tube's wall. The initial number of chemically active species embedded on the tube's inner surface  $(N_{\rm B})$  is comparable to the total number of A molecules  $(N_{\rm A})$  in the tube. This total initial number of A molecules can be expressed as  $N_{\rm A} = \int_{\rm tube} [A]_o dV$ , with dV being a differential volume in the AMS tube, and where  $[A]_o$  represents the initial and uniform concentration of A in the tube before binding begins. The total number of initial binding sites  $N_{\rm B}$  is,

likewise,  $N_{\rm B} = \oint_{\rm tube} [B]_o da$ , with da being a differential area on the AMS tube's inner wall.

For this discussion it is assumed that species A, B and the solvent molecules interact via standard intermolecular forces (e.g., polar electrostatic, hydrogen bonding, van der Waals).

A unique equilibrium will arise inside the AMS membrane and, by extension, in the two reservoirs. The equilibrium constant  $K_{eq}$  for the reaction in Eq. (1) is<sup>14,15</sup>:

$$K_{\rm eq} \equiv \frac{[AB]}{[A][B]} = \exp\left[\frac{-\Delta G}{kT}\right] = e^{-\beta\Delta G},\qquad(2)$$

where  $\Delta G$  is the Gibbs energy change for the reaction (J);  $k = 1.38 \times 10^{-23}$  J/K is Boltzmann's constant; T is the absolute temperature (K); thus, kT is the thermal energy and  $\beta = 1/kT$ . Here, [A] is the volume concentration of species A (molecules/m<sup>3</sup>), while [B] and [AB] are surface concentrations of species B and complex AB (molecules/m<sup>2</sup>). These concentrations are normalized appropriately (e.g., against one molar concentrations), thus rendering  $K_{\rm eq}$  dimensionless.

The initial distribution of open binding sites before reacting is  $[B(z)]_o$  and, after binding to some A molecules, it is reduced to [B(z)]. (Note: At equilibrium,  $[B(z)]_o > [B(z)]$ .) A useful expression for [A(z)] can be derived from Eq. (2) by noting that [AB] and [B] are linked: for every AB complex formed, one B site is lost, that is, d[AB] = -d[B]. One has, therefore,  $[AB(z)] = [B(z)]_o - [B(z)]$  With this, Eq. (2) can be recast as:

$$A(z)] = \exp\left[\frac{\Delta G(z)}{kT}\right] \left(\frac{[B(z)]_o}{[B(z)]} - 1\right) = \frac{1}{K_{eq}(z)} \left(\frac{[B(z)]_o}{[B(z)]} - 1\right).$$
 (3)

In principle, the following quantities in Eq. (3) can vary axially (along coordinate z), thereby making [A] a function of z: (i) the initial surface number density of B,  $[B]_o$ ; (ii) the surface number density of B evolving toward equilibrium [B]; and (iii) the local liquid-surface equilibrium constant,  $K_{eq}$  (or equivalently the free energy of binding,  $\Delta G$ ). Assuming all vary axially, the variation of [A] along the length of the tube can be written:

$$\frac{d[A(z)]}{dz} = - [A(z)]\frac{d}{dz} \left[\ln(K_{eq}(z))\right] + [A(z)]\frac{d}{dz} \left[\ln\left(\frac{[B(z)]_o}{[B(z)]} - 1\right)\right], \quad (4)$$

or equivalently in terms of Gibbs energy:

$$\frac{d[A(z)]}{dz} = [A(z)] \left[ \beta \frac{d(\Delta G(z))}{dz} \right] + [A(z)] \frac{d}{dz} \left[ \ln(\frac{[B(z)]_o}{[B(z)]} - 1) \right].$$
(5)

Here  $\frac{d[A]}{dz}$  is the *equilibrium* concentration gradient for the AMS. When properly engaged, it powers the AMCC.

Usually, equilibrium is the state in which concentration gradients are absent, but this is not the case for the AMS; rather, here a concentration gradient *is* the condition of thermodynamic equilibrium.

Equations (4,5) indicate that  $\left(\frac{d[A]}{dz}\right)$  can be induced either by varying the areal number density of binding sites  $\left(\frac{d[B(z)]_o}{dz}\right)$  or by varying the equilibrium constant  $\left(\frac{dK_{eq}(z)}{dz}\right)$  (or equivalently, the Gibbs energy,  $\Delta G(z)$ ). The former can be accomplished by locally seeding the walls with B appropriately at the start, while the latter can be achieved by grading the physical-chemical characteristics of species B itself. For example, if A is the hydrogen ion  $H^+$  and B is a carboxylate ion (COO<sup>-</sup>), then the binding strength of  $H^+$  to  $COO^-$  can be adjusted by various means. For instance, if COO<sup>-</sup> is attached to a carbon skeleton, functional groups can be attached at specific locations along the carbon framework to alter its acidity. As an illustration, consider the increase in pKa for the first three carboxylic acids: formic (one carbon; pKa = 3.75; acetic (two carbons; pKa = 4.74); propionic (three carbons; pKa = 4.87). This corresponds to a variation in  $[H^+]$  by factor of 13 between formic and propionic acids. By adding other functional groups nearby (e.g., alcohol, ketone, amine), the carboxylate's pKa can be further tailored along the length of the tube.

A hallmark of thermodynamic equilibrium is balance between competing processes, reactions, and forces. The equilibrium concentration gradient  $\frac{d[A(z)]}{dz}$  is no different. If the  $K_{eq}$  for the surface reaction (Eq. (1)) increases monotonically along the tube from left to right (Fig. 1b), then the first term on the rhs of Eq. (4) is negative definite, signaling larger [A] on the left than on the right (Fig. 1b). Likewise, if  $[B(z)]_o/[B(z)]$  increases monotonically from left to right, then the second term on the rhs of Eq. (4) is positive. (This can also be inferred from Le Chatelier's principle applied to the reaction.) Together, these countervailing tendencies limit the concentration gradient.

Once a concentration gradient has been established by the AMS, electrodes and an electrical load can be engaged to form a concentration cell, thus completing the AMCC. (See example in Appendix C.) The emf for a concentration cell ( $\mathcal{E}(V)$ ) is given by the Nernst equation:

$$\mathcal{E} = \frac{kT}{pq} \ln \left( \frac{a_h[A]_h}{a_l[A]_l} \right). \tag{6}$$

Here  $\mathcal{E}$  is the emf (V); p is the transfer number (the number of electrons transfered in the reaction, here taken as p = 1); q is an electronic charge  $(1.6 \times 10^{-19} \text{C})$ ;  $[A]_{h,(l)}$  is the high (low) concentration of A, and  $a_{h,(l)}$  is the chemical activity coefficient for the high (low) concentration (unitless). For dilute species,  $a \simeq 1$ . At T = 300 K,  $\mathcal{E}$  increases roughly 59 mV per decade difference between  $[A]_{h,l}$ , and the thermal voltage is kT/q = 26 mV.

A few items of note:

1) The concentration [A(z)] pertains to volume, while [B(z)],  $[B(z)]_o$ , [AB(z)], and G(z) pertain to the

surrounding surfaces. As such, this is a boundary value problem, in which the specifics of [A(z)] and  $\frac{d[A(z)]}{dz}$  can depend heavily on the size and shape of the reaction vessel (membrane channels). Numerical simulations bear this out<sup>2</sup>.

2) Analysis leading to Eqs. (3-5) does not explicitly include the effects of Brownian and Fick's diffusion of species A along the tube, both of which should attenuate [A(z)] and  $\frac{d[A(z)]}{dz}$ . This attenuation has been observed in numerical simulations of the AMS<sup>2</sup> and is corroborated by experiments<sup>1</sup>.

3) While Eqs. (4,5) indicate that the magnitude of  $\frac{d[A]}{dz}$  is proportional to [A], numerically large gradients do not necessarily translate to larger emfs<sup>16</sup> because the Nernst relation (Eq. (6)) dictates that it is the ratio  $\frac{[A]_h}{[A]_l}$  that determines emf, not their individual magnitudes.

#### B. Maximizing AMS Density Gradients

The density gradient  $\frac{d[A]}{dz}$  is the sine qua non of the AMS and AMCC. Strategies for maximizing it can be inferred by considering various operating limits of the AMS indicated by Eqs. (2-5).

- Limit 1: Trivially, if the AMS tube has initially uniform axial concentration of B (i.e.,  $\frac{d[B(z)]}{dz} = 0$ ) and also uniform binding strength (i.e.,  $\frac{dK_{eq}(z)}{dz} = 0$ ), then the concentration of A in solution will also be uniform (i.e.,  $\frac{d[A(z)]}{dz} = 0$ ), thus the AMS fails. Conclusion: At least one of the parameters,  $[B]_o$  or  $K_{eq}$  must vary axially.
- Limit 2: If the A-B binding is too strong, such that effectively  $K_{eq} \to \infty$ , then, assuming  $\frac{dK_{eq}(z)}{dz}$  remains finite, the first term on the rhs of Eq. (4) goes to zero, while the second term becomes irrelevant because A-B binding is effectively permanent, in which case there is no ongoing interplay between the walls and solute, thus A diffuses to uniform concentration  $(\frac{d[A(z)]}{dz} = 0)$ , and again, the AMS fails. Conclusion: Very strong surface binding should be avoided.
- Limit 3: In the opposite limit (very weak, effectively no surface binding), the walls are operationally inert, and therefore lack any chemical asymmetry, so [AB(z)] = 0 and  $K_{eq}(z) = 0$  along the tube. As a result, A molecules diffuse freely, uniformly filling the tube, rendering  $\frac{d[A(z)]}{dz} = 0$ Conclusion: Weak binding should likewise be avoided.
- Limit 4: If  $N_B \gg N_A$ , then regardless of the reactivity of A for B, the concentration of B will not change appreciably  $([B(z)] \simeq [B(z)]_o)$ , in which

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case Eq. (3) gives  $\lim_{[B]\to[B]_o}[A] = 0$ . In this scenario, species A has been stripped from the solution, precipitated out as AB, thus, the AMS fails. **Conclusion:**  $N_B \gg N_A$  should be avoided.

• Limit 5: In the opposite limit, if  $N_B \ll N_A$ , then the initial concentration of A is effectively unchanged (i.e.,  $[A(z)] \simeq [A]_o$ ). Without sculpting by [B(z)], the concentration of A will become uniform along the length of the tube via diffusion, in which case  $\frac{d[A]}{dz} = 0$ , and again the AMS fails. Conclusion:  $N_B \ll N_A$  should be avoided.

Clearly, the success of the AMS calls for moderation, as implied by the limiting conditions not conducive to it. Specifically, the limits to be avoided are:

i)  $\frac{d[B]}{dz} = 0$ ii)  $\frac{dK_{eq}}{dz} = 0$ iii)  $K_{eq} \to \infty$ iv)  $K_{eq} \to 0$ v)  $N_B \gg N_A$ vi)  $N_B \ll N_A$ 

These criteria follow naturally from Le Chatelier's principle and also comport with Sabatier's principle for catalysts<sup>17,18</sup>.

Further criteria for the AMS can be inferred, these involving temporal and spatial conditions. A necessary condition for non-zero  $\nabla[A]$  and  $\Delta[A]$  is that species A come into *local* chemical equilibrium with the tube walls either: (a) on an equilibration time scale  $\tau_{eq}$  short compared with A's diffusion time down an appreciable length of the AMS tube  $(\tau_{D,L} \sim \frac{L^2}{D}$  for Brownian diffusion, with  $\mathcal{D}$  being the diffusion coefficient for A in solution (m<sup>2</sup>/s)); or equivalently, (b) on a distance scale short compared to an appreciable length of the tube  $(L)^{19}$ . The chemical equilibration time  $(\tau_{eq})$  is the time required for A molecules to diffuse to the walls and to come into equilibrium with them and their binding sites B. The diffusion time  $\tau_{D,w}$  is the time necessary for A in the bulk solution to diffuse the diameter of the tube  $(\tau_{D,w} \sim \frac{w^2}{D})$ . Although just a few wall collisions may suffice to complete an AB reaction, chemical equilibrium cannot be consummated until a steady-state is reached between adsorption and desorption, which suggests that  $\tau_{eq}$  must be at least as long as A's residence time<sup>20</sup> on the binding site B,  $\tau_{res}$ . Furthermore, species A should reside on the surface long enough for diffusion along the tube to differentiate between regions of differing residence times; separation by perhaps a few w might suffice. In this case, a condition for non-zero  $\nabla[A]$  or  $\Delta[A]$  becomes  $(\tau_{D,w} \leq \tau_{\rm res})$ . The original condition (a) meanwhile, requires  $(\tau_{D,L} \gg \tau_{\rm res})$ . Putting them together, one has  $\tau_{D,w} \leq \tau_{\rm res} \ll \tau_{D,L}$  and  $w \ll L$ , the latter indicating that tubes should have large aspect ratios.

The limiting conditions (iii-vi) suggest that neither A nor B should chemically dominate the other and that, to some degree, they should be comparable in number  $(N_{\rm A} \sim N_{\rm B})$ , perhaps not equal but maybe within a few

orders of magnitude of each other<sup>21–24</sup>. Furthermore, if large  $\Delta A$  is sought, a large initial  $[A]_o$  is indicated and this, in turn, points to relatively large surface-to-volume ratios for the tubes such that  $N_B = \oint [B] da \sim \int [A] dV =$  $N_A$ . For this criterion, bundles of long, microscopic- or nanoscopic-diameter tubes are optimal.

# C. Dimensionless Constants and the Buckingham Pi Theorem

The behavior of the AMS tube is governed largely by the following six independent variables:

1)  $N_A$ : total number of A molecules in the AMS tube. (Dimensions: None.)

2)  $[B(z)]_o$ : initial density distribution of binding sites B as a function of axial location (z) along the tube walls. (Dimensions: m<sup>-2</sup>.)

3)  $\tau_{\rm res}(z)$ : residence time of A on binding site B as a function of axial location (z) along the tube. Here  $\tau_{\rm res}$  appears as a proxy for  $\Delta G(z)$  or  $K_{eq}(z)$ . (Dimensions: s.)

4) w: diameter of AMS tube. (Dimensions: m.)

5) L: axial length of AMS tube. (Dimension: m.)

6)  $\mathcal{D}$ : Diffusion coefficient for species A in solution. (Dimensions: m<sup>2</sup>/s) Typical value:  $\mathcal{D} \simeq 10^{-9} \text{ m}^2/\text{s}$  for small molecules and ions in water.

The AMS possesses three characteristic time scales  $(\tau_{D,w}, \tau_{D,L}, \text{ and } \tau_{\text{res}})$ , two characteristic length scales (*w* and *L*), and three characteristic particle numbers (*N*<sub>A</sub>, *N*<sub>B</sub>, and *N*<sub>AB</sub>), only two of which are independent (i.e.,  $dN_{AB} = -dN_A = -dN_B$ ). Two of the characteristic time scales are linked to the length scales: i.e.,  $\tau_{D,w} \simeq \frac{w^2}{D}$  and  $\tau_{D,L} \simeq \frac{L^2}{D}$ .

The number of dimensionless variables that define this system are set by the Buckingham Pi theorem  $(BPt)^{25}$ . Because there are six independent variables  $(N_A, [B(z)]_o,$  $\tau_{\rm res}(z), w, L, and D$  expressed in terms of two dimensional units (m, s), the BPt predicts 4 dimensionless variables. The tube aspect ratio, has already been identified:  $\alpha \equiv \frac{L}{w} \gg 1$ , based on considerations of diffusion times. Residence time  $(\tau_{\rm res})$  is the only independent variable solely involving time, so another time-dependent term must be invoked to create a dimensionless variable; here,  $\mathcal{D}$  is the natural choice, given the central role of diffusion in this system. As discussed earlier,  $\tau_{\rm res}$  cannot be inordinately long, but must be long enough to have thermodynamic effect. Also, the progression of characteristic times should be roughly:  $\tau_{D,w} \leq \tau_{res} \ll \tau_{D,L}$ . This suggests two dimensionless variables satisfying inequalities:  $\beta_w \equiv \tau_{\rm res}/\tau_{D,w} \ge 1$  and  $\beta_L \equiv \tau_{\rm res}/\tau_{D,L} \ll 1$ .

Finally, as discussed above, species A and B should compete with, but not dominate, each another, thus they should be somewhat comparable. This suggests a third dimensionless variable  $\gamma$ :

$$\gamma \equiv \frac{N_B}{N_A} = \frac{\oint [B]_o da}{\int [A]_o dV},\tag{7}$$

or area and volu

where the integrations over the interior area and volume of the AMS tube. With these, the AMS's operational regime can be circumscribed with the following four dimensionless parameters:

1)  $\alpha \gg 1$ ; 2)  $\beta_w \ge 1$ ; 3)  $\beta_L \ll 1$ ; and 4)  $0 < \gamma < \infty$ .

In addition to framing the operational limits of the AMS, these inform the numerical model of the membrane<sup>2</sup>.

#### III. DISCUSSION

The AMS is closely related to several well known chemical, electrochemical, and solid state systems. Starting at the molecular level, the AMS membrane can be understood as an extended, macroscopic version of a diprotic acid that has functional groups with disparate acidities. Just as a diprotic acid can display different acidities at opposite ends of the same molecule (e.g., linear perfluoropentadecane with a carboxylic acid group at one end and a sulfonic acid group at the other), likewise an AMS membrane, by virtue of its construction and composition, can also be made more acidic on one side than on the other; that is, it can present disparate chemical activities on opposite sides of the same structure<sup>27</sup> and, thereby, create a concentration gradient<sup>1</sup>.

# A. Liquid Chromatography and Concentration Gradient Corrosion

The chemical principles undergirding the AMS have well established precedents. The wide variety of liquidbased chromatographies<sup>28,29</sup>, for example, attest to the variability and controllability of surface binding site densities, binding energies, and residence times. In fact, an AMS tube operates similarly to a standard chromatographic column, with the following caveats: (a) binding sites B are inside the membrane rather than on the surfaces of beads in a column; (b) [B] and  $\Delta G$  are graded axially rather than being constant; (c) the desired product ( $\nabla[A]$ ) is a static, equilibrium fluid, rather a nonequilibrium fluid flow; and (d) the AMS equilibrium is achieved passively via diffusion.

The AMCC can also be understood in terms of one of the most ubiquitous and  $costly^{30}$  types of corrosion: *concentration gradient corrosion* (CGC)<sup>31–33</sup>, which commonly arises in aqueous environments with faying metal surfaces. For example, beneath an angle joint bolted to a metal base, dissolved metal ions are often concentrated compared with adjacent areas. This metal ion concentration gradient over a single piece of metal constitues a concentration cell, one that corrodes the metal in contact with the high ion concentration and deposits metal in the low concentration region. (CGC can also arise from gradients in dissolved atmospheric oxygen rather than from metal ions.) In our metal-centric civilization we are surrounded by electrochemical corrosion, most of it unwanted, unregulated, and unprofitable – and a fair bit of it CGC.

The AMCC is a controlled and useful version of CGC. The AMS purposefully establishes a concentration gradient, while the AMCC electrodes undergo electrochemical reactions that relax the gradient, specifically, the anode protects and the cathode corrodes. Their differences, however, are noteworthy. First, whereas the CGC is uncontrolled, the AMS has well defined current-voltage characteristics and mass transfer; it can be engineered and tuned. Second, and more importantly, whereas CGC is undesirable, ecologically destructive and economically costly, the AMCC is ecologically benign and potentially valuable.

The AMCC should demonstrate several advantages over other types of concentration cells – and perhaps even over some voltaics. First, the AMCC's  $\Delta[A]$  is selfgenerated and internal, rather than imposed by external means that require work input and chemical replenishment. This should simplify its operation, reduce support apparatus, and improve its overall efficiency. Second, if the AMS solutions are removed to power a concentration cell then returned to the AMS, their  $\Delta[A]$ s should return to their original values because this is the equilibrium state of the system. This  $\Delta[A]$  recurrence is spontaneous, mediated by thermal diffusion through the AMS membrane. In effect, with respect to its working solutions, the AMCC is self-charging.

To be clear, although the AMS solution is rechargeable, the full AMCC is not. As noted earlier, the AMCC is a case of controlled corrosion (CGC). As in the rudimentary example of Appendix C, its electrodes do not reconstitute themselves; rather the anode continually precipitates  $Cl^-$  as AgCl, accruing mass, while the cathode continually loses mass. As a result, the AMCC's operation does not constitute a closed thermodynamic cycle and does not undercut the second law of thermodynamics<sup>34</sup>.

The net mass transfer between electrodes ultimately exhausts the free energy of the AMCC, unless its electrodes are occasionally swapped. However, even without this maneuver the AMCC might offer increased service life and economy over traditional concentration cells. If it is fitted with an oversized cathode, then unlike conventional cells, which must be refilled after discharge, the AMCC should run multiple discharge-recharge cycles without refueling solutions, until the cathode is consumed. Its economy and compactness would thereby be improved because it requires less support apparatus for recharging, e.g., external circuits, plumbing, storage tanks.

A fundamental aspect of the AMS and its charging capability is that concentration cells operate at very low – at literally *thermal* – emfs. (Recall that at T = 300K,  $\frac{kT}{q} \simeq 26$ mV.) Along the entire length of an AMS  $\Delta G$ might reach only 4-5 kT of energy. Additionally, thermal energy drives diffusion in fluids and membranes, and it is also sufficient to support solvation reactions like acid association-dissociation. Because it is ultimately powered by ambient thermal energy, the AMCC might be called a *thermal battery* as well as a *concentration cell*.

#### B. Analogy to Solid State Diodes

AMS operation is analogous to that of solid state  $p-p^+$  diodes. The correspondences between *bipolar membranes* (BPMs)<sup>35-41</sup>, ion exchange membranes and traditional solid state n- and p-doped semiconductors, diodes and transistors have been noted by others<sup>42</sup>. In fact, membranes and solutions have been fabricated into *ionic diodes* and *ionic transistors* and even *ionic circuits*<sup>43-48</sup>. The AMS and AMCC are natural extensions of these. A full account is beyond the scope of this paper, however, a few introductory remarks are instructive.

The built-in potential across a standard *heterogeneous* p-n semiconductor diode  $(V_{\rm bi})$  is given approximately by the semiconductor version of the Nernst relation<sup>49-51</sup>:

$$V_{\rm bi,heter} = \frac{kT}{q} \ln\left(\frac{n_{\rm a}n_{\rm d}}{n_i^2}\right),\tag{8}$$

where  $n_{\rm a/d}$  (m<sup>-3</sup>) are the heterogeneous acceptor/donor concentrations and  $n_i$  is the intrinsic carrier concentration of the semiconductor (in silicon at 300 K,  $n_i \simeq 10^{16}$ m<sup>-3</sup>). For a representative case ( $n_{\rm a} = 10^{21}$  m<sup>-3</sup> and  $n_{\rm d} = 10^{24}$  m<sup>-3</sup> in silicon), one has  $V_{\rm bi,heter} = 0.8$  V.

Though often not appreciated, semiconductor diodes can also be fabricated *homogeneously*, that is, made completely from p-pure or n-pure materials – analogously to how an AMS employs a single chemical species. In this case, the built-in potential is given by:

$$V_{\rm bi,homog} = \frac{kT}{q} \ln\left(\frac{n_{\rm high}}{n_{\rm low}}\right),\tag{9}$$

where  $n_{\text{high/low}}(\text{m}^{-3})$  are the high/low concentrations of the single species. (Note the similarity between Eqs. (6) and (9).) A purely p-doped silicon diode with identical concentrations as the above silicon p-n diode gives  $V_{\text{bi,homog}} = 0.2 \text{ V}$  for the p-p<sup>+</sup> diode. Notice, too, the similar reduction of potential when switching from heterogeneous to homogeneous diodes (0.8 V versus 0.2 V), which mirrors the difference in  $\mathcal{E}$  between typical voltaic and concentration cells.

This analogy runs deeper. The depletion region of the p-p<sup>+</sup> diode is an electrophysical analog of an AMS membrane. Whereas the diode's depletion region and its  $V_{\rm bi}$  are the results of a difference in chemical potential and concentration between the p and p<sup>+</sup> regions (carrier reservoirs) relaxing to chemical equilibrium, in the AMS the chemical potential difference is built into the membrane itself, which then relaxes to equilibrium by generating concentration imbalances between the two thin solution reservoirs at their ends. The Nernst equation describes both systems<sup>49</sup>. Effectively, the AMS and p-p<sup>+</sup> diode are inside-out versions (inversions) of each other. In the p-p<sup>+</sup> diode, the species concentration differences of the bulk (i.e.,  $n_{\rm p}({\rm high}) > n_{\rm p}({\rm low})$ ) determines the character of the interface (depletion region), whereas in the AMS, it is the inverse: the character of the interface (membrane) determines the species concentration differences in the bulk ( $n_{\rm A}({\rm high}) > n_{\rm A}({\rm low})$ ).

The spontaneous recharging of the AMS is also analogous to the spontaneous recharging in semiconductor systems involving pn junctions. Both are driven by gradients in chemical potentials and are mediated by diffusion. It is well known that an equilibrium-state electric field resides at the junction between p-doped and n-doped regions of static semiconductors (the so-called depletion region) and is due to the thermal cross-diffusion of p- and n-type charge carriers. These fields can be modified by perturbing the diode's electronic or mechanical boundary  $conditions^{52,53}$ . However, when these perturbations are removed, the electric fields revert to their original values because, after all, they represent the equilibrium state for the original configuration of semiconductors. (This reversion is rapid, often taking just 10-100 ns.) Likewise, when the  $\Delta[A]$ s are expended in the AMCC's concentration cell, they are re-established when returned to the AMS. The implications of this are intriguing, particularly as they pertain to the second  $law^{52,53}$ .

#### C. Energy and Power Density

Energy and power densities are standard metrics for battery performance. Energy density  $\rho_{\rm e}({\rm J}/{\rm m}^3)$  indicates how much work can be done by a battery of a given volume, whereas power density  $\mathcal{P}(W/m^3)$  indicates how fast that amout of work can be done. In terms of  $\rho_{\rm e}$ , the AMCC, like other types of concentration cell, should be inferior to standard voltaics by roughly 2-3 orders of magnitude; however, its power densities might be comparable to that of commercial voltaics (e.g.,  $\mathcal{P}(\text{lithium-}$ ion) $\simeq 10^6 W/m^3$ ). An estimate of AMCC power density  $(\mathcal{P})$  can be made as follows. Assume  $\mathcal{P} \sim \frac{\mathcal{E}I}{V_{\text{tube}}}$ , where  $\mathcal{E}$  is cell emf given by the Nernst relation, Eq. (6); I is the particle diffusion current in an AMS tube  $(I = qJ_{\text{diff}} \cdot A_{\text{tube}}) \simeq q(\mathcal{D}\nabla[A]) \cdot w^2; A_{\text{tube}} \simeq w^2$  is the cross-sectional area of the tube; and  $V_{\text{tube}} = L \cdot w^2$  is the tube volume (Fig. 1). Here  $J_{\text{diff}} = -\mathcal{D}\nabla[A]$  is the particle diffusion current density (particles/ $m^2$  s), as per Fick's law<sup>54,55</sup>.

Invoking the Einstein relation  $(\mathcal{D}/\mu = kT/q)$ , where  $\mu$  (C·s/kg) is the charge mobility, the AMCC power density  $\mathcal{P}(W/m^3)$  can be shown to scale as:

$$\mathcal{P} \sim kT \ln\left(\frac{[A]_h}{[A]_l}\right) \frac{\mathcal{D}[A]}{L^2} = \mu[A] \left(\frac{kT}{L}\right)^2 \ln\left(\frac{[A]_h}{[A]_l}\right).$$
(10)

The last form is revealing: thermal energy (kT) and size (L) appear quadratically. Both the Nernst emf and the diffusion current contribute a kT term; clearly, this sys-

tem is thermally driven. Small device length L, high species mobility  $\mu$ , elevated temperatures, and high concentration ([A]) also favor large  $\mathcal{P}$ . Of these, L is probably the parameter most amenable to engineering improvements: thin membranes and reservoirs are recommended.

As a concrete example of  $\mathcal{P}$ , assume the hydrochloric acid AMCC above<sup>1</sup>, with pH = 1, a  $\Delta$ pH = 1 (concentration factor of 10), the diffusion coefficient of aqueous chloride ions ( $\mathcal{D} \simeq 10^{-9} \text{ m}^2/\text{s}$ ), and cell size ( $L = 5 \times 10^{-6}\text{m}$ ). With these, Eq. (10) predicts a maximum power density in excess of  $10^7 \text{ W/m}^3$ . Standard inefficiencies in fluid and heat transfer, however, as well as electrochemical nonidealities, should reduce this number significantly. Engineering issues surrounding this will be considered in future investigations.

Energy densities  $(J/m^3)$  for a single-discharge AMCC should be inferior to those of standard voltaics by 2-3 orders of magnitude ( $\rho_e$ (Lithium-ion) ~ 2 × 10<sup>9</sup>J/m<sup>3</sup>). However, because the AMCC is self-charging, its *inte*grated energy density (energy summed over all dischargerecharge cycles) should be superior to other types of concentration cells and, if it can complete  $10^2 - 10^3$  thermochemical cycles before its electrodes are exhausted, it might be comparable to commercial voltaics.

It will be interesting to see whether the AMS selfcharging capability can be transplanted to voltaic cells with their larger emfs and energy densities. While this cannot yet be ruled out, there are good theoretical reasons to doubt it. The thermal energy density of a material (excluding phase changes) scales as  $\rho_{\rm e}$  (thermal) ~ nkT, where n is the particle number density (m<sup>-3</sup>). For T = 300 K, the thermal energy of a molecule is roughly a few  $kT \simeq 4 \times 10^{-21} \text{J} \simeq 1/40 \text{eV}$ . (Here, vibrational modes are ignored because most are inactive at room temperature.) The chemical energy density of a material (say TNT) scales as  $\rho_{\rm e}$  (chemical) ~ n\Delta E, where  $\Delta E$  the reaction energy. For the explosive decomposition of TNT,  $\Delta E \simeq 1.6 \times 10^{-18} \text{J} \simeq 10 \text{ eV}$  per molecule. The ratio of thermal to chemical energy density, therefore, scales as:  $\frac{\rho_{\rm e}(\text{thermal})}{\rho_{\rm e}(\text{chemical})} = \frac{kT}{\Delta E} \simeq 10^{-2} - 10^{-3}$  for typical physical systems. Because the AMCC is powered by the AMS, which itself is driven by thermal diffusion, the upgrade of thermal energy to that of chemical reactions requires an energy concentration process. No such process is currently known; moreover, it would be prohibited by the second law. Thus, for the moment, AMCC energy density seems limited to its thermal energy density, which is inherently  $10^2 - 10^3$  times less than that of traditional voltaic sources. However, given that AMS rechargeability allows for multiple thermal energy helpings – subject, of course, to other constraints like electrode consumption – this limit seems a soft one.

The AMS effect is quite general and should extend beyond electrochemical applications. Given a suitable membrane, in principle, the AMS should be able to separate or concentrate many species of interest, perhaps aiding such processes as the desalination of seawater<sup>56,57</sup> or the recovery of metals from waste water streams<sup>58</sup>. Again, note that AMS-mediated separation is energyneutral, passively relying on molecular diffusion, thus on the thermal energy of its environment rather than on an external free energy source as is required, say, by reverse osmosis in desalination<sup>56,57</sup>.

# IV. CONCLUSIONS AND FUTURE DIRECTIONS

A new type of electrochemical cell is proposed, the *asymmetric membrane concentration cell* (AMCC). Like other concentration cells, the AMCC exhibits relatively low energy density compared to most voltaics, but it has advantages, namely, its simplicity, rechargeability and economy of design.

The novel feature of the AMCC is the *asymmetric membrane separator* (AMS), which generates the concentration gradient for the AMCC. In this study it is modeled as a bundle of identical, long, thin microscopic tubes with chemical potential gradients built into their walls. Its equilibrium concentration gradient can be understood in various ways:

- Traditional thermodynamics predicts the AMS effect. Analysing  $K_{eq}$  for the surface reaction (A + B  $\rightleftharpoons$  AB) identifies the primary factors upon which [A(z)] depends, specifically, the initial binding site density  $([B]_o)$  and the Gibbs free energy  $(\Delta G)$ , in Eqs. (3-5). Because [B] and  $\Delta G$  can be engineered to vary with z, [A] must likewise vary ([A] = [A(z)]).
- The AMS effect can be seen to arise as a balance between oppositely directed particle current densities in the membrane  $(J_{\text{diff}} \text{ vs. } J_{\nabla G})$ . These generate a one-dimensional profile for [A(z)] akin to that of an isothermal atmosphere (Eq. (11)) in Appendix A.
- Equilibrium solute concentrations in the AMS tube can be modeled in 1-3 dimensions using the timeindependent (equilibrium) diffusion equation (Appendix B). The solution of the 2-D AMS, has a wellknown analog, the Laplace equation solution for the electrostatic parallel plate capacitor. From it one can deduce that a concentration gradient must form in the AMS.

A number of physical embodiments of the AMS and AMCC are possible, two of which are discussed in the companion article to this one<sup>1</sup> and one in Appendix C. High aspect-ratio AMSs with microscopic tubes holding high concentrations of mobile ions are expected to provide high device power density, which might exceed  $10^7$  Wm<sup>-3</sup>. Key virtues of the AMS are that it: (a) is self-contained and does not require external solution reservoirs; and (b) spontaneously recharges its concentration

difference  $\Delta[A]$ . Like other ionic solution systems, the AMS has analogs in solid state physics, notably, the n-p and p<sup>+</sup>-p diodes. The AMS concept might have utility in other arenas, such as desalination and metal recovery from waste streams.

Toy model numerical simulations of the membrane tube, currently in progress, corroborate the principal findings of this study<sup>2</sup>. The model simulates Brownian diffusion of individual solute molecules (A) in an AMS tube (§II), subject to the binding reaction: A + B  $\rightleftharpoons$ AB. The density of surface binding sites  $[B]_o$  and the average residence time of A on B ( $\tau_{\rm res}$ ) can be varied, along with other core variables (e.g., L, w, D). Simulated concentration gradients are in good qualitative agreement with theory. The long-term goal is to provide realistic quantitative data in order to optimize experimental and commercial membrane design<sup>59</sup>.

The present theoretical inquiry is far from complete. The AMS model considers only electrically neutral species A and B, while in real-world scenarios A and B are likely to be ionic, in which case solute A will be accompanied by counter-ions, as is the case with laboratory AMCCs<sup>1</sup>. Ions complicate the model by introducing electrostatic and electrochemical phenomena such as the Debye layer, zeta potential, *electrical double layer* (EDL) and the diffuse layer.

While electrostatic effects will complicate the AMS model, they should not affect its primary outcomes. First, the thermodynamic analysis of §II.A is still germane and clearly predicts an equilibrium concentration gradient. Second, electrokinetic effects should be minor or nil because the AMS/AMCC does not involve physical fluid flow; it is diffusion driven. Third, for the concentrations envisioned for this model – and explored in laboratory experiments<sup>1</sup> – the EDL and Debye layer should be narrow compared with the radii of the AMS channels over most of their range. Here the channel diameters are taken to be  $(10^{-8} \text{m} \le w \le 10^{-6} \text{m})$ , whereas for 1 molar hydrochloric acid the Debye length is expected to be less than or on the order of  $10^{-8}$ m. The details of electrostatic and electrochemical effects are beyond the scope of this paper but will be taken up in future studies.

Additional investigation of the physical chemistry of asymmetric membrane-solution interactions seem warranted. For historical comparison, the ubiquitous membrane material Nafion remains a focus of theory and experiment even now, more than 50 years since its discovery, and it is probably a simpler chemical system than the AMS membrane. With the optimal design parameters for the AMCC unknown, numerical simulations might be decisive. Further inquiry into the connections between the AMS and semiconductors, chromatography, and corrosion might also be fruitful.

Lastly, an intriguing question remains as to whether the AMCC electrodes can be reconstituted in a thermodynamically spontaneous fashion; if so, this could have far-reaching implications for the foundations of thermodynamics and for energy sustainability. The chemically active solutions in the AMCC should be recyclable indefinitely because solutes are not destroyed, merely relocated. If the electrodes can be similarly recycled, then the AMCC would constitute a chemically-closed but thermally-open system, one powered solely by local ambient thermal energy. This would be a boon for sustainable energy and it would open new vistas for thermodynamics<sup>60</sup>.

In summary, the AMCC is a new type of concentration cell and its self-charging capability is a new concept in battery design whose applications could extend beyond electrochemical arena into such arenas as desalination and waste stream recovery. It appears many theoretical and experimental surprises lie ahead. Transcending these academic concerns, however, is the hope that the AMCC will contribute to the urgent worldwide effort for sustainable energy.

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- <sup>63</sup>The solid endcaps can be replaced with chemically treated fine screens. This would preserve the mathematical boundary conditions, while potentially allowing the flow of solution into and out of the cell.
- <sup>64</sup>Consider the following AMS test case:  $L \simeq 10^{-5} \text{m} \gg w \simeq 10^{-7} \text{m}$ ) and a concentration gain of 10 along the tube. The two criteria  $(L \gg w)$  and [A] varying slowly enough with z to model the parallel plate capacitor are both reasonably well satisfied.
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#### **Appendix A: Equilibrium Current Densities**

The asymmetric membrane concentration cell (AMCC) – or any concentration cell, for that matter – requires two reservoirs at distinct concentrations in contact with its anodic and cathodic electrodes. Creating a spatial gradient in the solute concentration is the fundamental requirement; it is the arete of the AMS (asymmetric membrane separator). It was shown in §II using traditional thermodynamics that this gradient can arise naturally as the equilibrium state in an AMS tube.

In this appendix, it is shown that this equilibrium state can also be viewed as the balance between oppositely directed one-dimensional current densities of species A within the AMS tube, one due to particle diffusion, the other due to the gradient in the chemical potential imposed by the asymmetric wall composition<sup>61,62</sup>.

Diffusion current density, written from Fick's law, is:  $J_{\text{diff}} = -\mathcal{D}\nabla[A(z)] = -\mathcal{D}\frac{d[A(z)]}{dz}$ , where  $\mathcal{D}$  is the diffusion constant for A. The drift current density, driven by a chemical potential gradient, can be written:  $J_{\nabla G} \simeq -[A(z)]\mu\nabla(G(z))$ , where G(z) is the axially varying Gibbs free energy per reaction (J), and  $\mu$  is the mobility of A. (For neutral A the units of mobility are (s/kg).)

The diffusion and  $\nabla G$  currents are oppositely directed and at equilibrium they balance each other. Setting them equal  $(J_{\text{diff}} = J_{\nabla G})$  and appealing to the Einstein relation  $(\mathcal{D}/\mu = kT)$ , one obtains:

$$[A(z)] = [A(0)] \exp\left[\frac{\Delta G}{kT}\right].$$
 (11)

Here  $\Delta G = G(z) - G(0)$  and  $[A(0)] \equiv [A(z = 0)]$ . Following the left-right convention in Fig. 1, and because  $\Delta G < 0$  for the binding reaction (A + B  $\rightleftharpoons$  AB), one has [A(z)] < [A(0)], indicating an equilibrium concentration difference between the ends of the tube, the defining feature of the AMS (Fig. 1b). (If one unpacks  $\Delta G$  differently, one can obtain Eqs. (2,3).)

This result is not surprising. Equation (11) is a chemical analog of an exemplar: the vertical distribution of molecules in an isothermal atmosphere in a uniform gravitational field, which is written:  $[A(z)] = [A]_{sl} \exp[\frac{-mgz}{kT}]$ . Here, *m* is the mass of an individual A gas molecule (kg); g is the gravitational acceleration  $(m/s^2)$ ; z is the vertical altitude; and  $[A]_{sl}$  is A's concentration at sea level. Clearly, the gravitational potential energy of A varies with altitude ( $\Delta U = -mgz$ ). In an AMS the chemical force on A (i.e.,  $\nabla G$ ) can be billions of times stronger than the gravitational force (i.e., mq), hence [A(z)] can change appreciably over sub-micron distances as apposed to over hundreds or thousands of meters in a planetary atmosphere. Additionally, the functional dependence of G(z) is more flexible than that of the gravitational potential. For instance, if G is constant, then  $\frac{\partial G}{\partial z} = 0$ , there is no gradient, therefore no chemical force, in which case diffusion erases all concentration gradients in the solution; thus, the AMS and AMCC fail. (In

the gravitational case, this corresponds to no gravitational field, in which case gas number density will be uniform.) If the Gibbs energy varies linearly with z (i.e.,  $G(z) = -\kappa z$ , with  $\kappa$  a constant (kg·m/s<sup>2</sup>), then [A] varies exponentially like an isothermal atmosphere  $([A(z)] = [A(0)] \exp(\frac{-\kappa z}{kT}) = [A(0)] \exp(-z/\lambda)$ , with an e-folding distance  $\lambda = \frac{kT}{\kappa}$ .

For a robust AMCC,  $\lambda$  should be sufficiently short that multiple e-foldings accrue along the length of the AMS tube so as to attain a large concentration difference in A, while still maintaining reasonably short diffusion times. (Recall that diffusion times  $t_D$  typically scale as the variance of the displacement:  $\langle x^2 \rangle \simeq 2nDt$ , where n is the system's dimensionality, n = 1, 2, or 3.) To illustrate, an increase in concentration by a factor of 100 along the tube requires that the binding energy of species A must decrease by roughly  $\Delta G \simeq (4.6)kT$  over that distance. This is can be arranged via  $[B(z)]_o$  or  $\tau_{\rm res}(z)$ , as discussed earlier (§II.A). Overall, in contrast to gravitational potential, G(z) can exhibit a good deal of functional flexibility consistent with an effective AMS.

# Appendix B: Time-Independent Diffusion Equation and AMS

In this appendix, the AMS concentration gradient is approached from the standpoint of diffusion of species A inside the membrane tube, with chemically active walls. The A-B surface reaction (Eq. (1)) is enforced as mathematical boundary conditions.

Up to this point, analysis of AMS equilibrium has been limited to one dimension (z); here it is extended to 2-D using the time-independent diffusion equation:

$$-\mathcal{D}\nabla^2[A(x,y,z)] = 0, \qquad (12)$$

where  $\nabla^2$  is the Laplacian operator. Equation (12) describes the time-independent thermodynamic equilibrium state for the concentration [A(x, y, z)] inside the AMS tube.

Referring to Fig. 2, let the AMS tube possess the same axial chemical asymmetry assumed previously (§II). The value of [A] at the boundaries can be calculated from Eq. (3) using G(x, z) and [B(x, z)], which are defined at the outset. The following physical symmetries simplify the analysis:

(1) Mirror (bilateral) symmetry across the z axis, i.e., f(x) = f(-x).

(2) No y-dependence for [A], thus  $\left(\frac{\partial^2[A]}{\partial^2 y} = 0\right)$ , in which case the Laplacian operator is reduced from 3-D to 2-D and the AMS assumes slot geometry. If y-dependence is desired for a full 3-D description, the 2-D tube might at least be made a square channel, thereby symmetrizing the x and y solutions. (Going forward, 2-D slot geometry is assumed.)

(3) The ends of the AMS (z = 0 and z = L) are chemically identical to their immediate lateral walls<sup>63</sup>.

Formally, the boundary conditions for the AMS in Fig. 2 are written:



FIG. 2. Specifications of AMS tube for the application of the diffusion equation (Eq. (12)). Blue represents the tube walls. In narrow channel geometries like this, solute concentration [A(x, z)] should closely follow the wall concentration distribution of binding sites, [B(z)].

(i) Bottom Boundary  $(x = -w/2; 0 \le z \le L)$ : [A(-w/2, z)] = f(z) (variable); (ii) Top Boundary  $(x = +w/2; 0 \le z \le L)$ : [A(+w/2, z)] = f(z) (variable); (iii) Endcap 1  $(z = 0; -w/2 \le x \le +w/2)$ :  $[A(x, 0)] = f(z = 0) = C_1$  (constant); and (iv) Endcap 2  $(z = L; -w/2 \le x \le +w/2)$ :  $[A(x, L)] = f(z = L) = C_2$  (constant)

With well-defined initial values for G(x, z),  $[B(x, z)]_o$ ,  $N_{A,o}$ , and T (temperature) the system must come to an equilibrium state characterized by unique values of [B(x, z)] and [AB(x, z)] on the walls, as well as a unique profile of [A(x, z)] in solution. The boundary conditions (i - iv) alone, however, are not sufficient to solve the diffusion equation. Additional local and global constraints must be imposed, namely, particle conservation relations. For these,  $\oint_{S,V}$  indicates a closed integral over system boundary surfaces (S) or volume (V); subscript (o) indicates initial value. These are informed by the limit conditions and dimensionless variables in §II B,C. **Local constraints:** 

(a) 
$$[B(x, z)] \leq [B_{o}(x, z)];$$
 and  
(b)  $[AB(x, z)] = [B_{o}(x, z)] - [B(x, z)].$ 

## **Global constraints:**

(c) 
$$\oint_{\mathbf{S}}[AB] \equiv N_{\mathbf{AB}} \leq N_{\mathbf{B},\mathbf{o}} = \oint_{\mathbf{S}}[B(x,z)]_{o}$$
  
and  
(d)  $\oint_{\mathbf{S}}[AB] \equiv N_{\mathbf{AB}} \leq N_{\mathbf{A},\mathbf{o}} = \oint_{\mathbf{V}}[A(x,z)]_{o}$ .

The time-independent diffusion equation for the AMS (Eq. (12)) might yield to separation of variable techniques, subject to the above physical boundary conditions and constraints. A full, analytic solution is beyond the scope of this paper and might be intractable except for simple boundary conditions (e.g., constant  $[B]_o$  and  $\Delta G$ ). This problem, however, seems well-suited to numerical solution, perhaps using relaxation methods, par-

tial differential equation solvers, or finite element methods like Comsol Multiphysics or ANSYS. We are currently modeling this system via individual-particle Brownian diffusion inside an AMS tube subject to chemically active walls<sup>2</sup>. Initial results corroborate the principal findings of this paper.

That Laplace's equation is involved  $(\nabla^2[A] = 0)$  is fortuitous because its solutions are guaranteed to satisfy the following three properties:

(1) The interior solution is uniquely determined by [A] on the boundary, in this case, the solution-surface interface. (2) For the 2-D case, the value of [A] at a spatial point (x,z) is the average of values on its surrounding circle:  $[A(x,z)] = \frac{1}{2\pi R} \oint_{\text{circle}} [A(x,z)] dl.$ (3) Solutions have no local maxima or minima in the

interior; all extrema are on the boundaries.

A complete solution to  $\mathcal{D}\nabla^2[A(x,z)] = 0$  is not provided here, but a familiar analog is known: the electrostatic parallel plate capacitor. The mathematical form of the time-independent diffusion equation for the high-aspect-ratio AMS slot  $(\nabla^2[A] = 0)$  is identical to that of the well-known, high-aspect-ratio plate capacitor (with electrically conducting endcaps) written as Laplace's equation:  $\nabla^2 V(x,z) = 0$ . A twist arises because the value of [A] in the AMS slot (tube) changes along the z direction, but for the capacitor this can be easily handled with Laplace's equation by varying the boundary values of V along z.

Physical intuition about the AMS tube can be gleaned from this analog case. For a parallel plate capacitor, so long as  $L \gg w$  and so long as V(z) varies slowly, the interior values of voltage closely match those on the boundary. By analogy, and because of the three properties of Laplace solutions stated  $above^{64}$ , we have a good approximation for the interior values of [A(x,z)], namely, its equilibrium values at the nearest solution-surface boundary, which are calculable from Eq. (3). Thus, if G(z) and  $[B(z)]_o$  are engineered with gradients, then gradients in [A(z)] must also obtain.

# Appendix C: Physical Example of AMCC

In this appendix, a physical instantiation of the AMCC is briefly described, one distinct from the laboratory version reported on elsewhere<sup>1</sup>.

Consider an AMS plumbed to a concentration cell inspired by the standard AgCl/Ag pH probe<sup>26,65</sup> (Fig. 2). The AMS is filled with hydrochloric acid (HCl), hence A is the hydrogen ion  $H^+$  and  $Cl^-$  the counter ion. The binding sites B might be carboxylate ions (COO<sup>-</sup>) attached to carbon skeletons anchored along the walls of the AMS in such a real number densities and with such functional groups attached that the hydrogen ion concentration  $[H^+(z)]$  decreases vertically, as depicted in Fig. 1b.

To maintain quasi-neutrality, the local Cl<sup>-</sup> concentration closely follows that of the hydrogen ion. Notice how the electrochemical roles of  $H^+$  and  $Cl^-$  reverse between



FIG. 3. Schematic of AMCC. (a) Full system. AMS (right side) connected by values and plumbing to concentration cell (left side). (b) Concentration cell magnified, with load resistor  $R_{\rm L}(\Omega)$  (not engaged). For the AMCC power cycle, the load  $R_{\rm L}$  is engaged, the V-1 values are open, and the V-2 valve is closed, thus admitting active solutions into concentration cell for electricity generation. When the solution is chemically exhausted, the V-1 valves are closed and the V-2 valve is opened so solutions can return to the AMS for reseparation. This cycle can be repeated until the electrodes in the concentration cell are exhausted or, if they are regularly switched such that the electrodes do not degrade, then, in principle, this cycle can be repeated indefinitely.

the AMS and the concentration cell. The AMS membrane explicitly concentrates H<sup>+</sup>, with Cl<sup>-</sup> coming along for the ride, while in the concentration cell,  $Cl^-$  is the electrochemically active species, with H<sup>+</sup> coming along for the ride.

Once the equilibrium hydrogen ion concentration is established in the AMS, values V-1 are opened and  $H^+$  is admitted to the concentration cell. The cell (Fig. 3b), consists of two Ag/AgCl electrodes separated by a semipermeable membrane, e.g., here Nafion, which is permeable to  $H^+$  but not to  $Cl^-$ . The anode's oxidation half-reaction (high  $[H^+]$ ) is:

$$\operatorname{Cl}^-_{(\operatorname{high})} + \operatorname{Ag}_{(s)} \longrightarrow \operatorname{AgCl}_{(s)} + \operatorname{e}^-_{:s}$$

while the cathode's reduction half-reaction  $(low [H^+])$  is:

$$\operatorname{AgCl}_{(s)} + e^{-} \longrightarrow \operatorname{Ag}_{(s)} + \operatorname{Cl}_{(\operatorname{low})}^{-}$$

Summing the two half-reactions, the full electrochemical reaction is:

$$\operatorname{Cl}^-_{(\operatorname{high})} \longrightarrow \operatorname{Cl}^-_{(\operatorname{low})},$$

which is emblematic of a concentration cell.

Clearly, the net electron transfer between species is nil; nonetheless, the concentration difference between the two solutions relaxes via electron current through the external load  $(R_{\rm L}(\Omega))$  and proton current through the central (nafion) membrane, which acts as the salt bridge.

Once the  $\Delta pH$  wanes, the values (V-1) are closed, V-2 is opened, and the expended solutions are returned to the AMS, where  $\Delta[H^+]$  is re-established in the AMS. As for traditional voltaic cells, AMCCs can be combined in series and parallel to boost emf and current<sup>66</sup>.

The first laboratory AMCCs relied on multi-layer, nafion-based custom membranes and achieved only a few percent concentration difference between the anode and cathode chambers, thus, generated only modest emfs ( $\mathcal{E} \sim 1 \text{mV}$ ). More recent experimental AMCCs utilize commercial anionic exchange membranes to separate low-molarity NaCl solutions. These generate larger emfs (e.g.,  $\sim 5 \text{ mV}$ ), indicating that the concentration differences between chambers might be as large as 15-20%.

The most recent laboratory membranes physically resemble the AMS membrane from §II.A (Fig. 1). Specifically, solid membrane material (thickness ~  $10^{-3}$ m) is drilled through with small holes (diameter ~  $10^{-4}$ m) in high areal number density (~  $10^3$  holes/cm<sup>2</sup>); thus, highaspect-ratio tubes ( $L/w \geq 10$ ) connect the anode and cathode fluid reservoirs.

By no means have these AMS membranes or AMCCs been optimized. It is expected that large series-parallel arrays of them will soon drive simple electronic appliances. These developments will be reported upon in future communications.